

## Hierarchical emergent structure in commercial colloidal and polymeric systems

11:30am - 12:45pm US / Canada - Eastern - August 24, 2021 | Room: Zoom Room 21

[Gregory Beaucage](#), Presenter; [Kabir Rishi](#)

Division: [PMSE] Division of Polymeric Materials Science and Engineering

Session Type: Oral - Virtual

In commercial products additives with limited or no miscibility with a base resin or solvent are often encountered. Through manipulation of interfacial agents and kinetic milling, mixing or extrusion, nano- to colloidal-scale additives assemble into micron- to millimeter-scale structures that can gel or produce desired optical scattering, electrical conduction, mechanical reinforcement, and engineered rheology. These emergent multi-hierarchical networks are important in many applications. We have studied several examples of these systems. Generally they can be categorized into thermally dispersed and mechanically dispersed systems. **Thermal dispersion:** Inks containing organic pigments are miscibilized in water using non-ionic surfactants. Depending on the phase behavior of these surfactants and the milling history, the primary structure, aggregate structure and macroscopic network structure can be tuned. In a printer the complex structure can be subject to shear that further influences the structure. Finally, as concentration increases with drying a complex, robust network on the micron scale, fused with binder in some cases, can scatter light and give mechanical strength to the dried film. Although the initial, shelf-stable system is thermally dispersed, a complex interplay between kinetic and thermodynamic features govern the resulting film. **Kinetic dispersion:** For reinforced elastomers and many filled polymers such as carbon black filled polyethylene for UV protection or polymer solid electrolytes the additives are inherently immiscible, yet it is desired to produce a macroscopic network from nanoscale structures for mechanical strength, electrical conductivity and optical and UV scattering. In these systems we can make a crude approximation that the accumulated strain shows similarity to temperature in thermally dispersed systems in that it leads to random dispersion of nano- or colloidal particles. We can make some gross approximations using what is known about thermally dispersed colloidal systems to understand the impact of excluded volume and interaction energies in the context of a kinetic pseudo-<sup>1</sup> temperature. This approach has proven fruitful in understanding dispersion with some nuances. For instance. in

# Hierarchical emergent structure in commercial colloidal and polymeric systems

**Greg Beaucage**, Professor of Chemical and Materials Engineering

**Kabir Rishi**, PhD Candidate

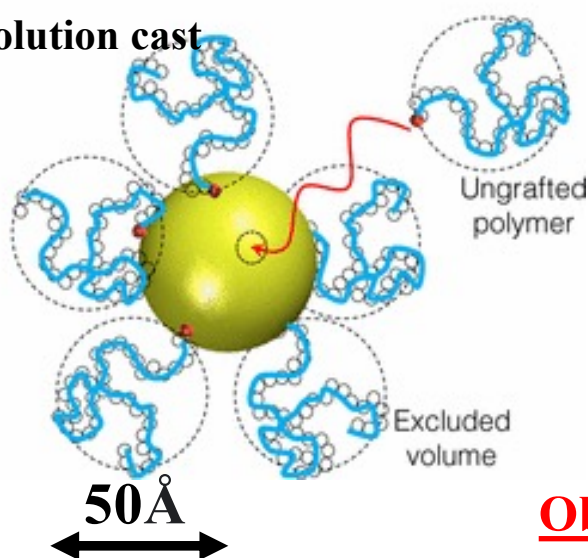
Department of Materials Science & Engineering, University of Cincinnati

# Current state-of-the-art nanocomposite

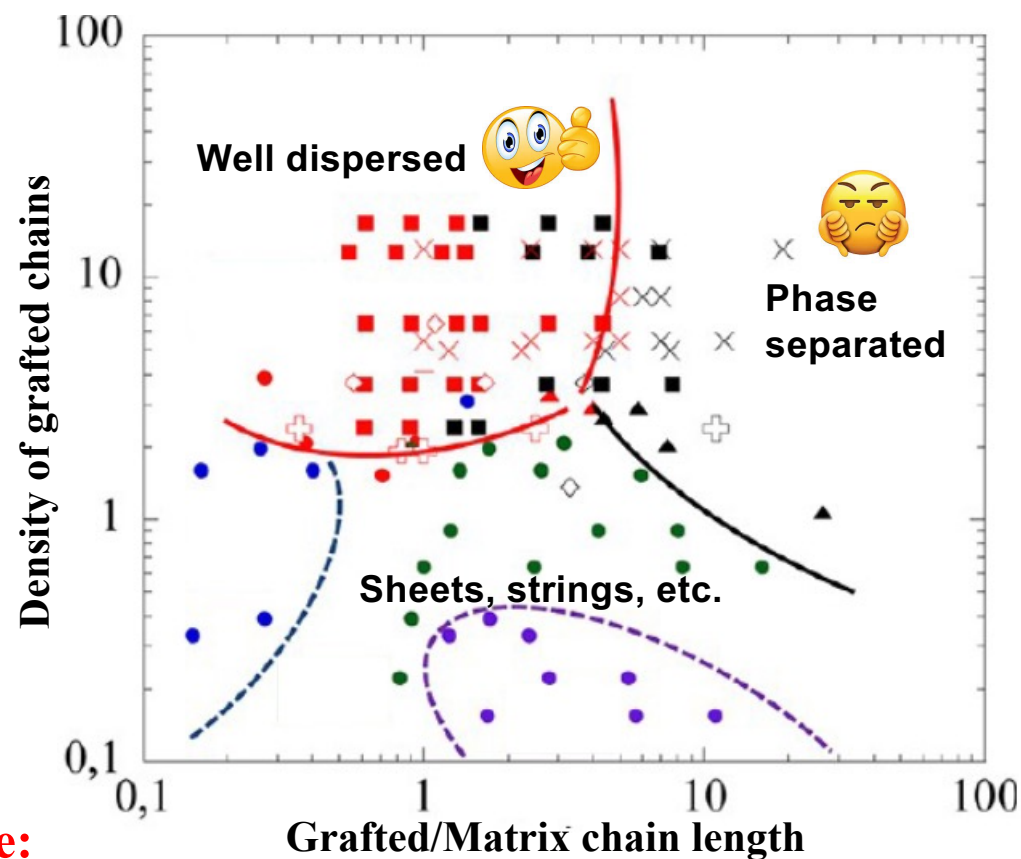
Monodisperse colloidal particles

Colloidal Solution cast

Thermally Dispersed



**Objective:**  
**Enhance miscibility**



Kumar, S.K., Jouault, N., Benicewicz, B. and Neely, T., 2013. Nanocomposites with polymer grafted nanoparticles. *Macromolecules*, 46(9), pp.3199-3214.

Kumar, S.K., Benicewicz, B.C., Vaia, R.A. and Winey, K.I., 2017. 50th anniversary perspective: are polymer nanocomposites practical for applications?. *Macromolecules*, 50(3), pp.714-731.

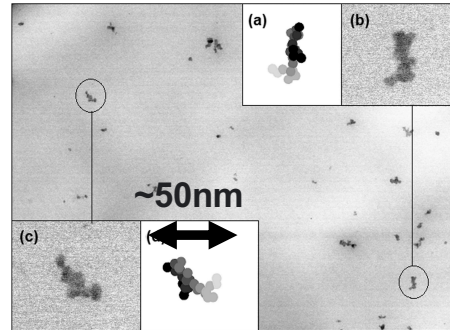
Asai, M., Zhao, D. and Kumar, S.K., 2017. Role of grafting mechanism on the polymer coverage and self-assembly of hairy nanoparticles. *ACS Nano*, 11(7), pp.7028-7035.

# The original nanocomposite

Polydisperse aggregates

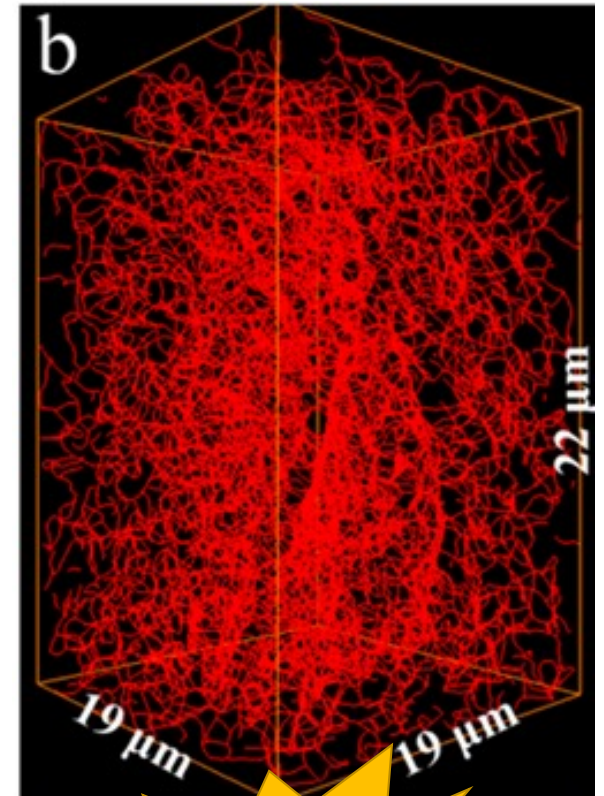
Processed under shear

Kinetically mixed **immiscible**



**Objective:**  
**Tear resistance**  
**Static charge dissipation**

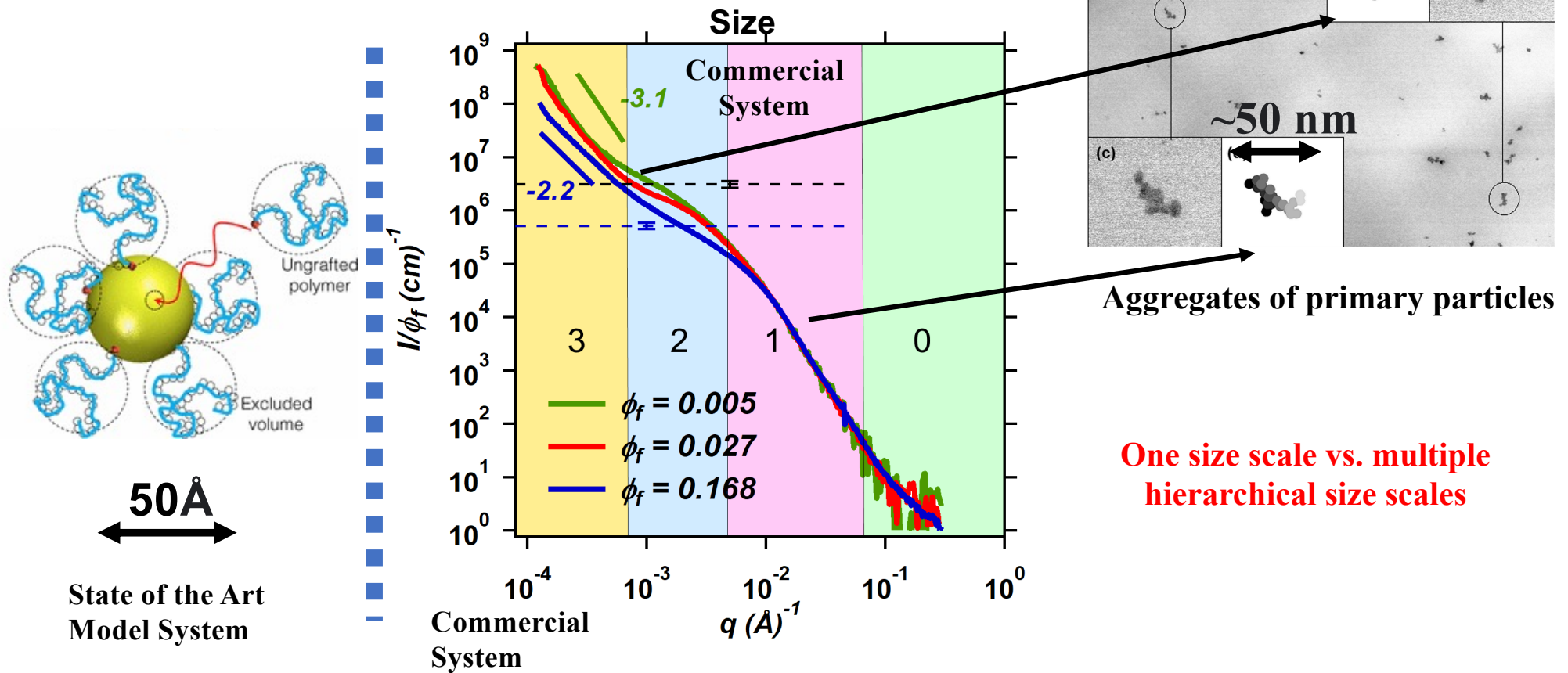
**Why/how do added nanoparticles impact structures 100-1,000 times larger?**



**1,000 x  
larger**

Song, L.; Wang, Z.; Tang, X.; Chen, L.; Chen, P.; Yuan, Q.; Li, L. Visualizing the Toughening Mechanism of Nanofiller with 3D X-ray Nano-CT: Stress-Induced Phase Separation of Silica Nanofiller and Silicone Polymer Double Networks Macromolecules 50 7249-7257 (2017).

# Multiscale Hierarchical Structures

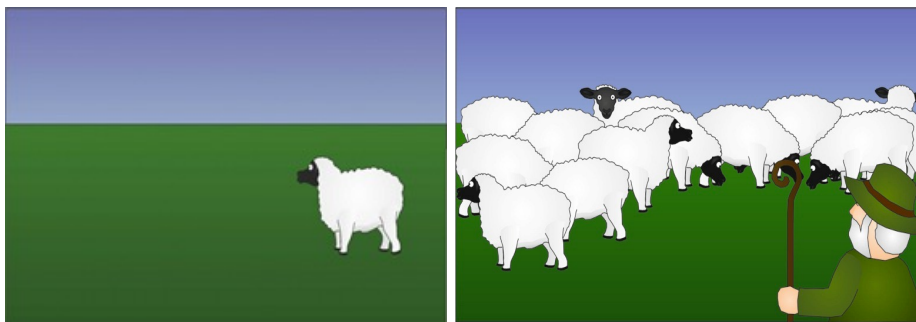


Rishi, K., Beaucage, G., Kuppa, V., Mulderig, A., Narayanan, V., McGlasson, A., Rackaitis, M. and Ilavsky, J., 2018. Impact of an emergent hierarchical filler network on nanocomposite dynamics. *Macromolecules*, 51(20), pp.7893-7904.

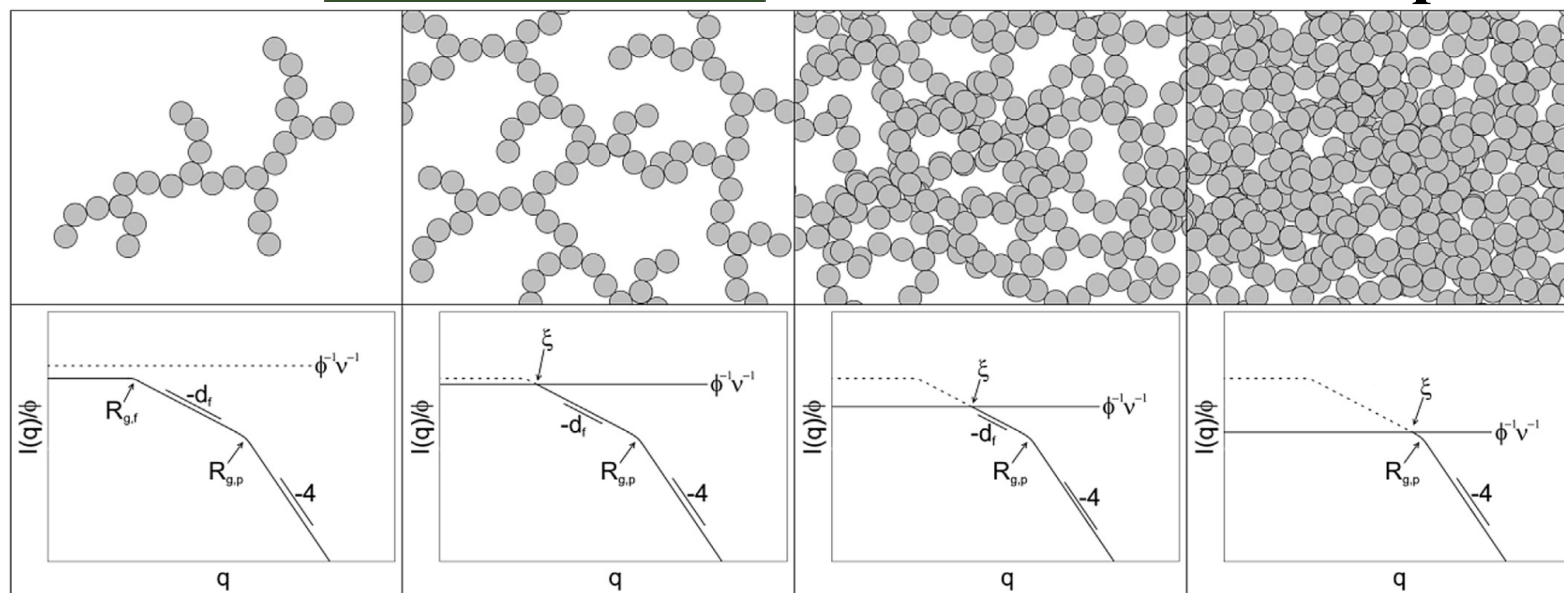
Mulderig, A., Beaucage, G., Vogtt, K., Jiang, H. and Kuppa, V., 2017. Quantification of branching in fumed silica. *Journal of Aerosol Science*, 109, pp.28-37.



# $A_2$ from scattering



# Quantitative measure of nano-dispersion



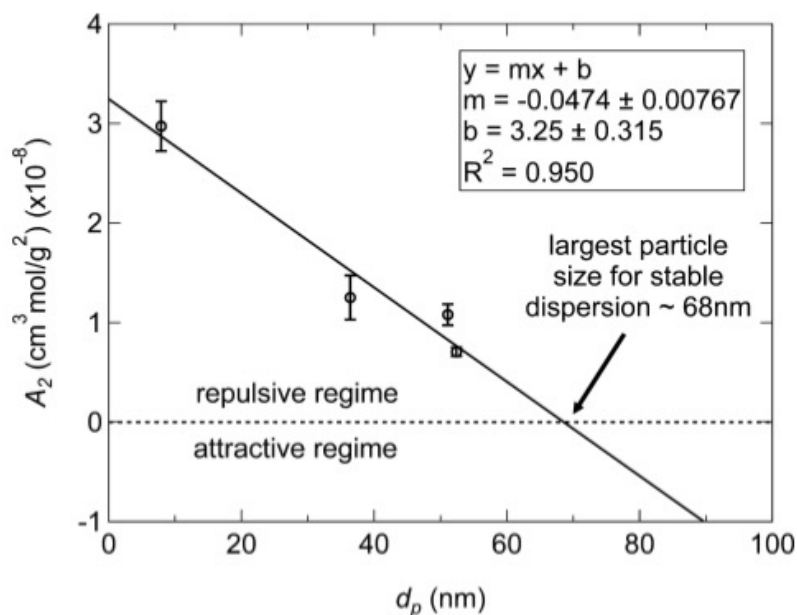
Pedersen, J. S.; Sommer, C. Temperature Dependence of the Virial Coefficients and the Chi Parameter in Semi-Dilute Solutions of PEG. In *Scattering Methods and the Properties of Polymer Materials*; Springer Berlin Heidelberg: Berlin, Heidelberg, 2005; pp 70–78.

Vogtt, K.; Beaucage, G.; Weaver, M.; Jiang, H. Thermodynamic Stability of Worm-like Micelle Solutions. *Soft Matter* 2017, 13 (36), 6068–6078.

Jin, Y.; Beaucage, G.; Vogtt, K.; Jiang, H.; Kuppa, V.; Kim, J.; Ilavsky, J.; Rackaitis, M.; Mulderig, A.; Rishi, K.; Narayanan, V. A Pseudo-Thermodynamic Description of Dispersion for Nanocomposites. *Polymer (Guildf)*. 2017, 129, 32–43.

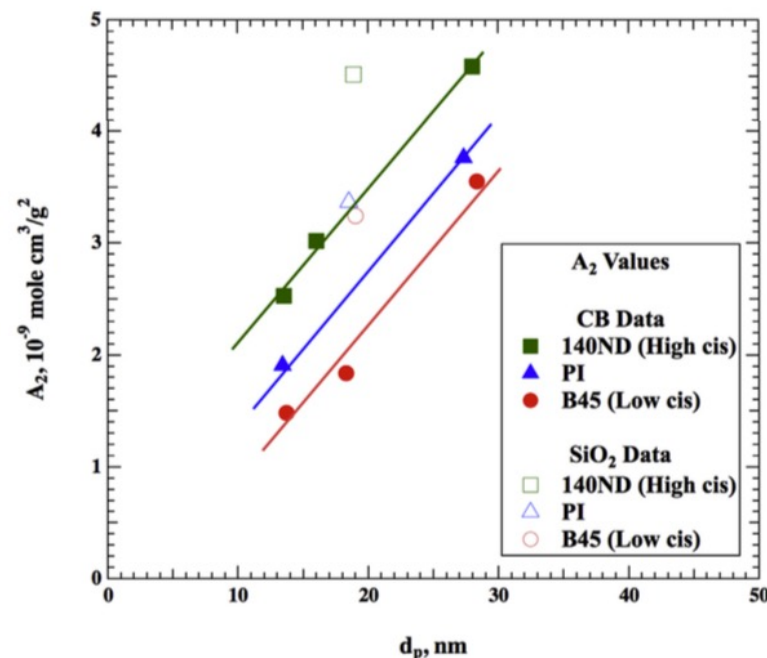
# Thermal Dispersion versus Kinetic Dispersion

## Miscible Organic Pigment with Triton X100



Thermally driven nano-dispersion /  
Stokes drag coefficient

## Immiscible Carbon Black and Silica in Elastomer



Mechanically driven nano-dispersion /  
Lever arm

Mulderig, A.; Beaucage, G.; Vogtt, K.; Jiang, H.; Jin, Y.; Clapp, L.; Henderson, D. C. Structural Emergence in Particle Dispersions. *Langmuir* **2017**, 33 (49), 14029–14037.  
Jin, Y.; Beaucage, G.; Vogtt, K.; Jiang, H.; Kuppa, V.; Kim, J.; Ilavsky, J.; Rackaitis, M.; Mulderig, A.; Rishi, K.; Narayanan, V. A Pseudo-Thermodynamic Description of Dispersion for Nanocomposites. *Polymer (Guildf)*. **2017**, 129, 32–43.

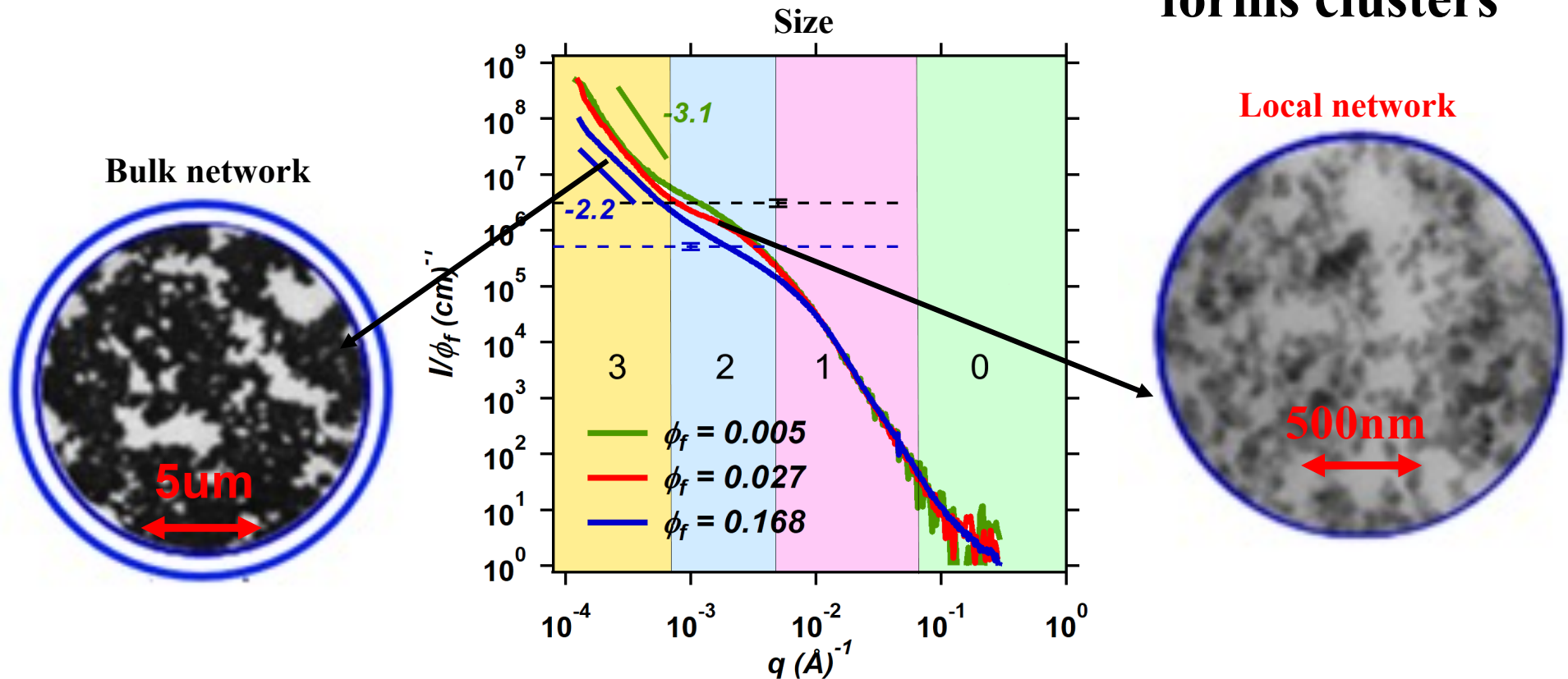
## Clustering can lead to locally higher concentrations





# Multiscale Hierarchical Structures

Immiscibility  
forms clusters

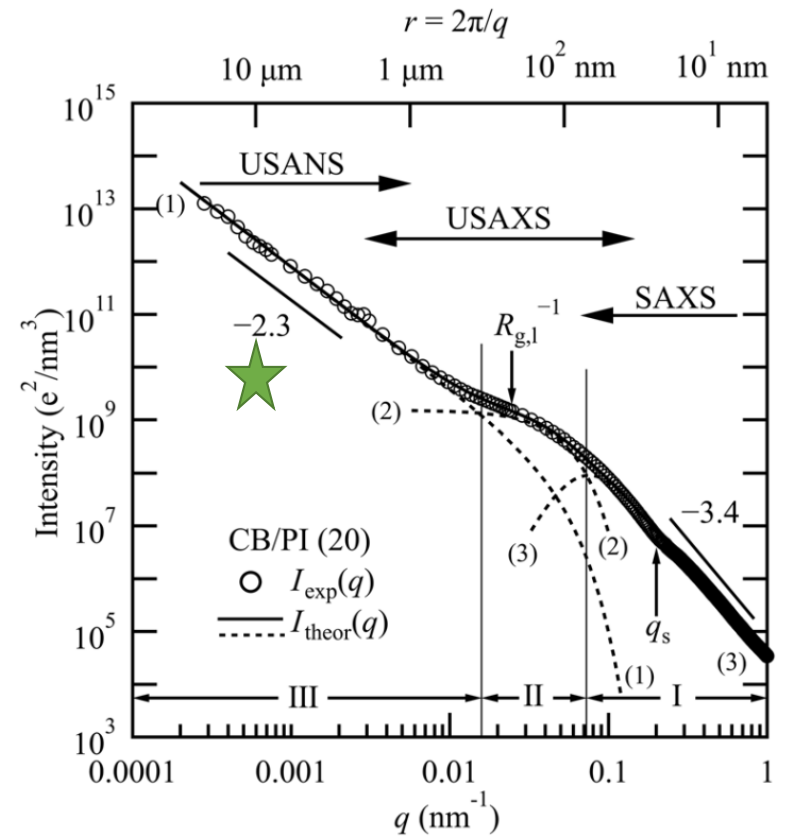
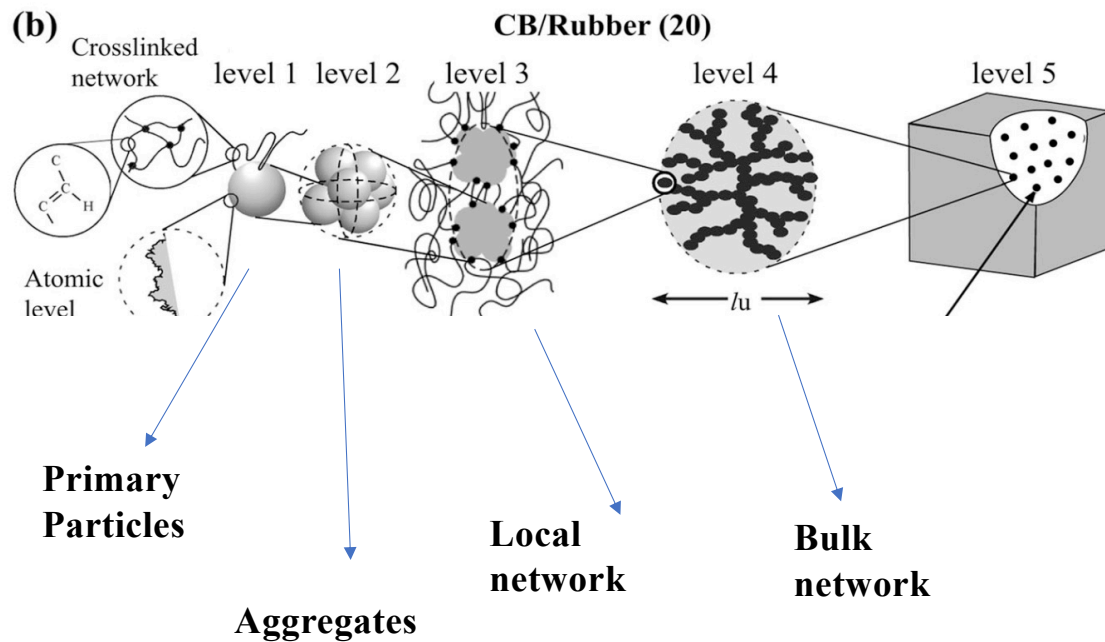


Rishi, K., Beaucage, G., Kuppa, V., Mulderig, A., Narayanan, V., McGlasson, A., Rackaitis, M. and Ilavsky, J., 2018. Impact of an emergent hierarchical filler network on nanocomposite dynamics. *Macromolecules*, 51(20), pp.7893-7904.

Trappe, V. and Weitz, D.A., 2000. Scaling of the viscoelasticity of weakly attractive particles. *Physical review letters*, 85(2), p.449.

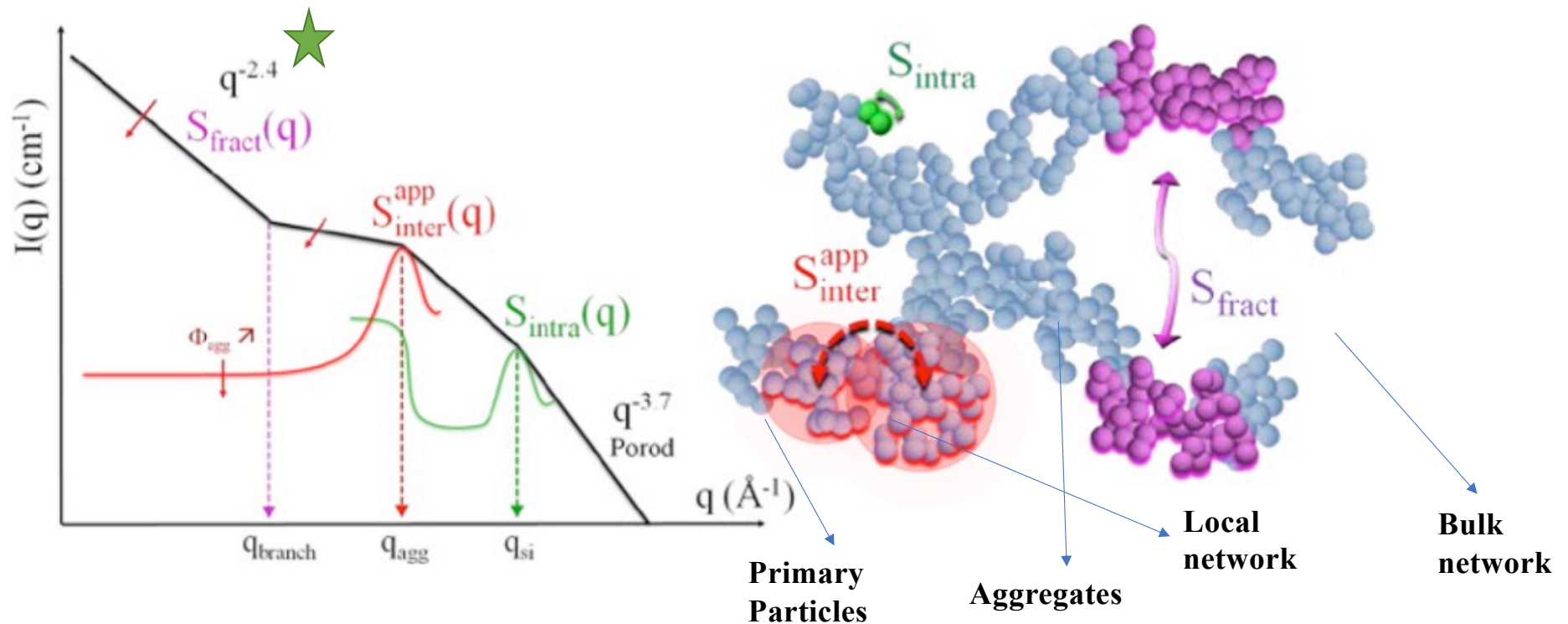
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# Multiscale Hierarchical Structures



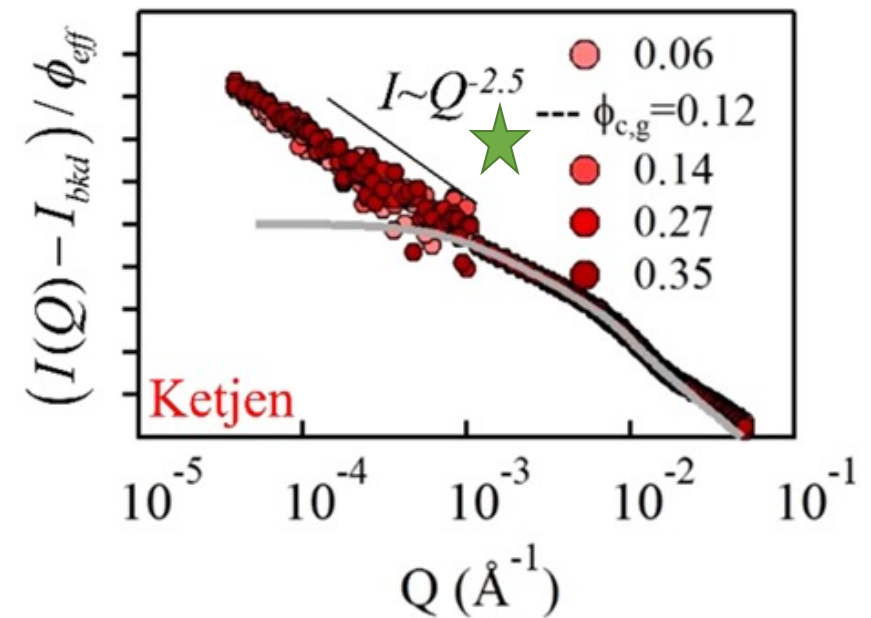
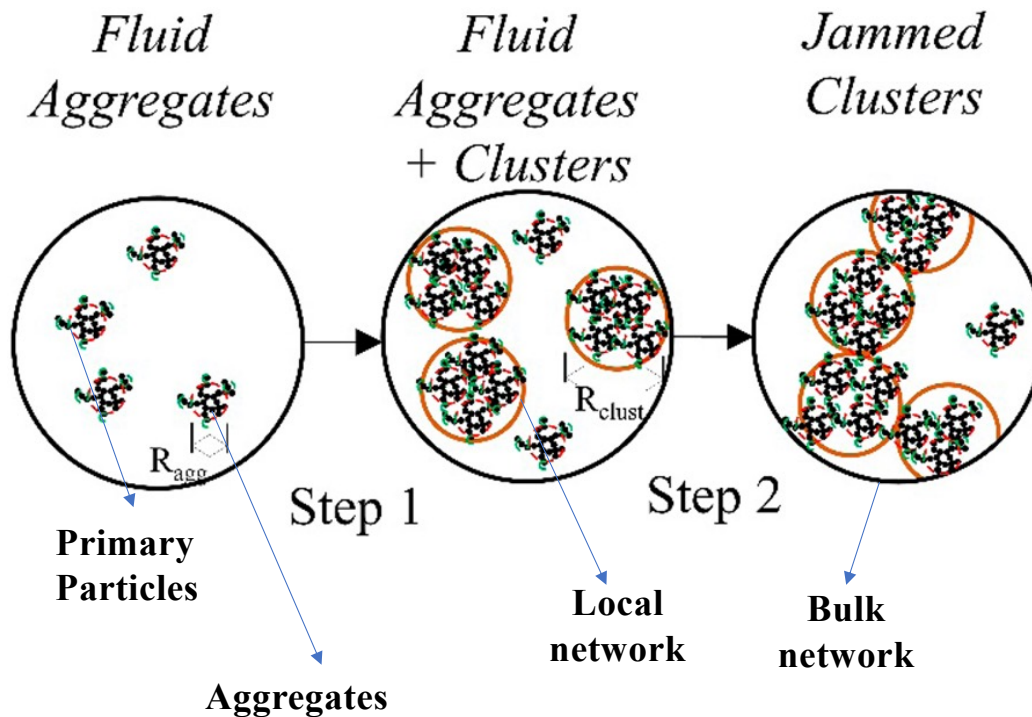
Hashimoto, T., Amino, N., Nishitsuji, S. and Takenaka, M., **2019**. Hierarchically self-organized filler particles in polymers: cascade evolution of dissipative structures to ordered structures. *Polymer Journal*, 51(2), pp.109-130.

# Multiscale Hierarchical Structures



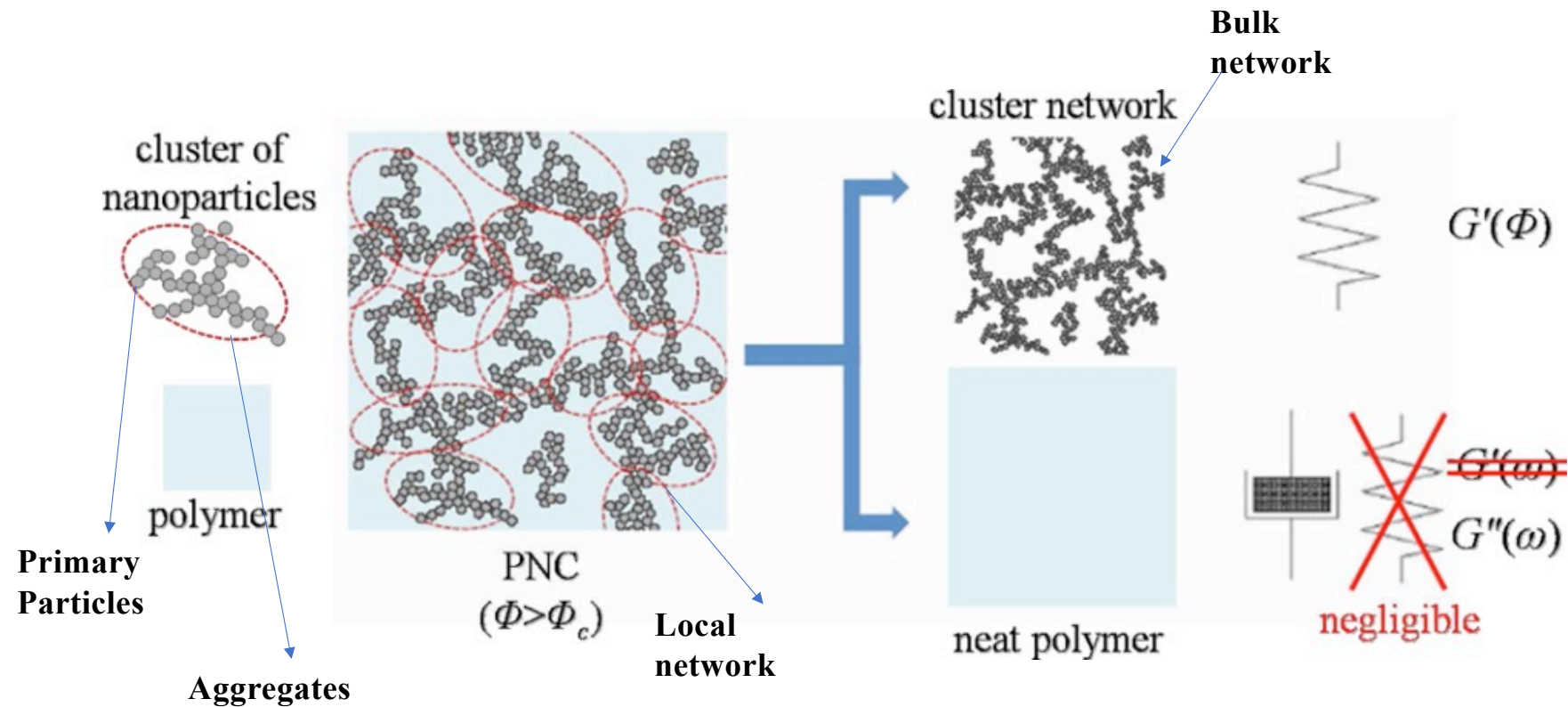
Baeza, G.P., Genix, A.C., Degrandcourt, C., Petitjean, L., Gummel, J., Couty, M. and Oberdisse, J., **2013**. Multiscale filler structure in simplified industrial nanocomposite silica/SBR systems studied by SAXS and TEM. *Macromolecules*, 46(1), pp.317-329.

# Multiscale Hierarchical Structures



Richards, J.J., Hipp, J.B., Riley, J.K., Wagner, N.J. and Butler, P.D., **2017**. Clustering and percolation in suspensions of carbon black. *Langmuir*, 33(43), pp.12260-12266.

# Multiscale Hierarchical Structures



Filippone, G., Romeo, G. and Acierno, D., **2010**. Viscoelasticity and structure of polystyrene/fumed silica nanocomposites: filler network and hydrodynamic contributions. *Langmuir*, 26(4), pp.2714-2720.

Filippone, G. and Salzano de Luna, M., **2012**. A unifying approach for the linear viscoelasticity of polymer nanocomposites. *Macromolecules*, 45(21), pp.8853-8860.



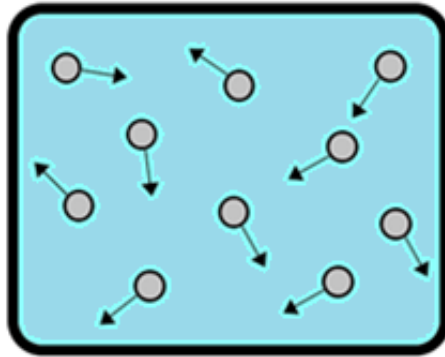
# van der Waals model for incompatible polymer nanocomposites

$a$  reflects the attractive energy of interaction between aggregates.

$$\Pi = \frac{RT}{(V + b)} - \frac{a}{V^2}$$

$b$  is the excluded volume

**Thermally driven colloidal dispersion**

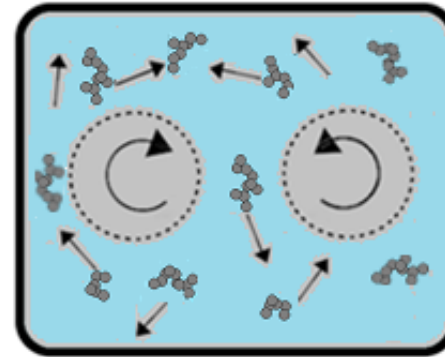


**Heat Source**

**Energy  $\propto$  Temperature**

$$A_2(T) = \frac{N_A}{M^2} \left( b - \frac{a}{kT} \right)$$

**Mechanically dispersed nano-fillers**

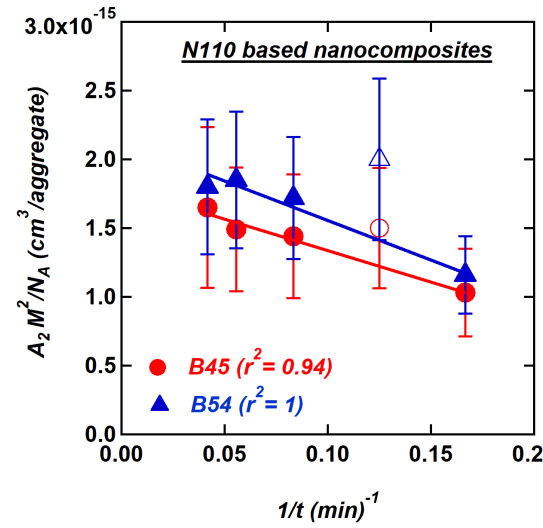


**Energy  $\propto$  Mixing Time**

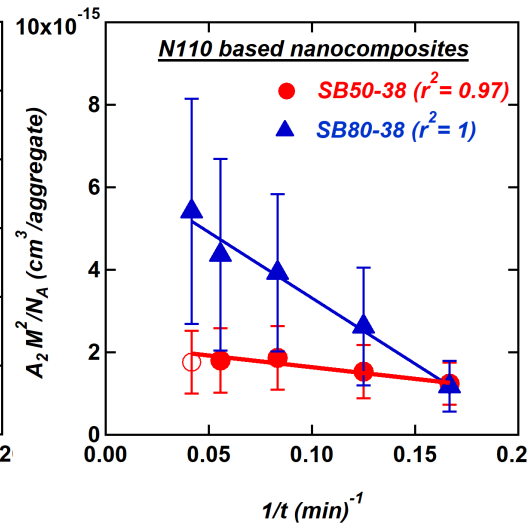
$$A_2(t) = \frac{N_A}{M^2} \left( b^* - \frac{a^*}{t} \right)$$

$\approx$

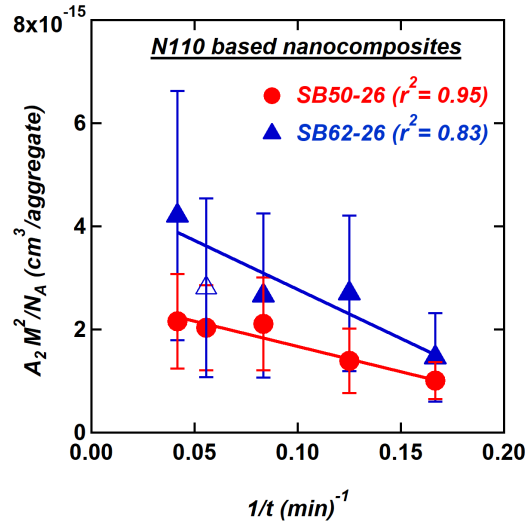
$$T \sim \langle \dot{\gamma} \rangle = Nt\psi$$



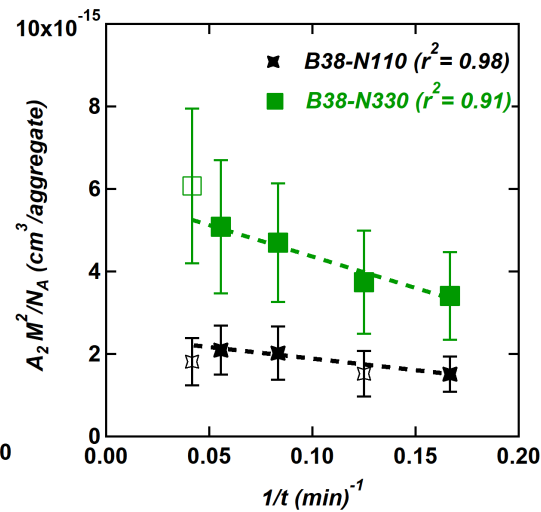
(a)



(b)



(c)



(d)

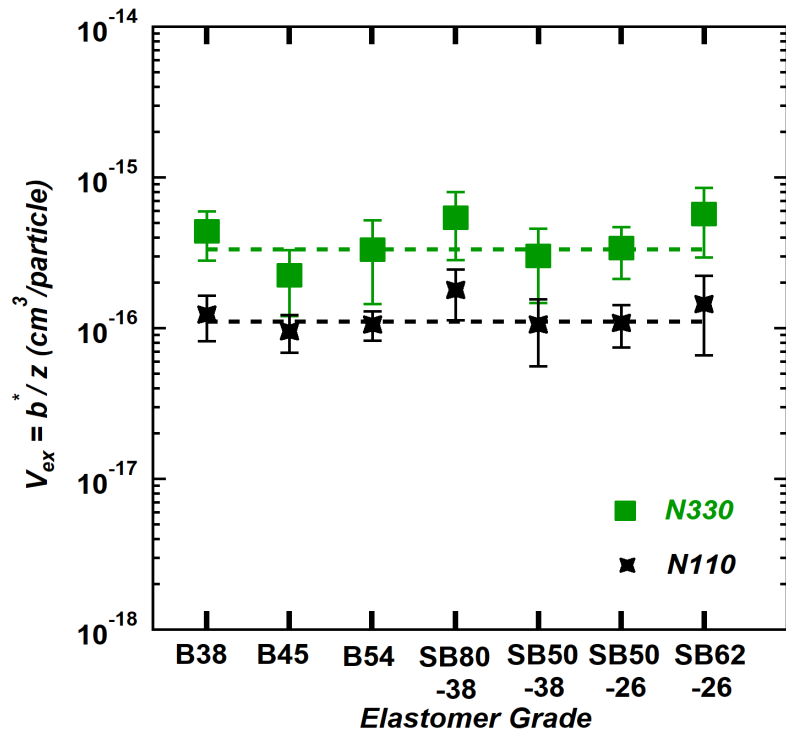
Van der Waals approach seems viable

$$A_2(t) = \frac{N_A}{M^2} \left( b^* - \frac{a^*}{t} \right)$$

Rishi, K.; Narayanan, V.; Beaucage, G.; McGlasson, A.; Kuppa, V.; Ilavsky, J.; Rackaitis, M. A Thermal Model to Describe Kinetic Dispersion in Rubber Nanocomposites: The Effect of Mixing Time on Dispersion. *Polymer (Guildf)*. 2019, 175, 272–282.

Excluded volume is associated with the occupied volume of an aggregate.

$$(d_{p,N330}/d_{p,N110})^3 = 4.2$$

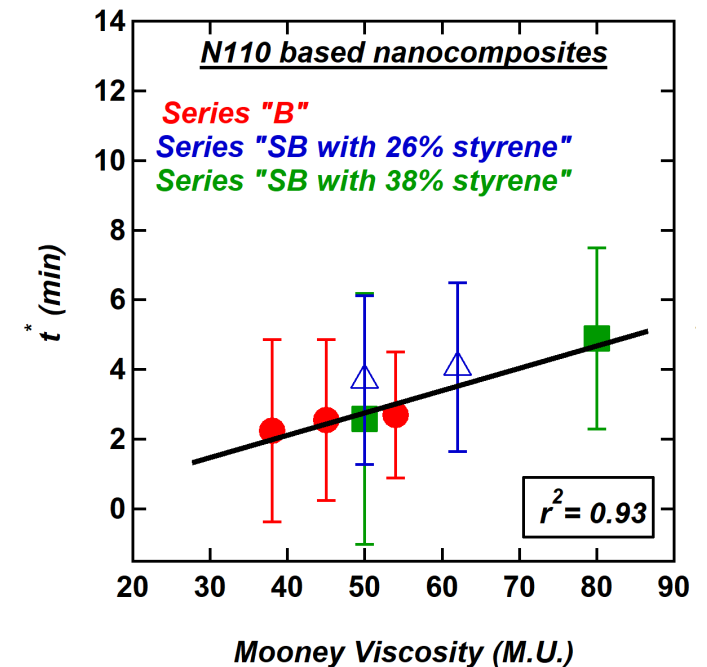


$$A_2(t) = \frac{N_A}{M^2} \left( b^* - \frac{a^*}{t} \right)$$

Wetting time depends on viscosity and primary particle size

x-intercept reflects “wetting time”

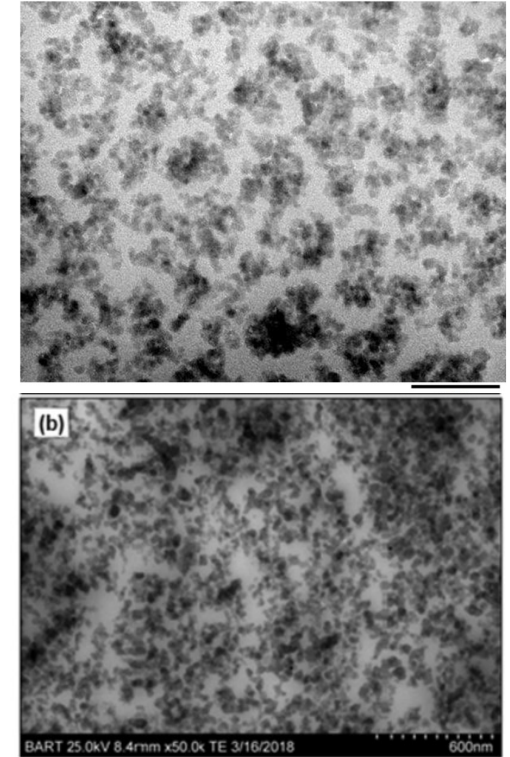
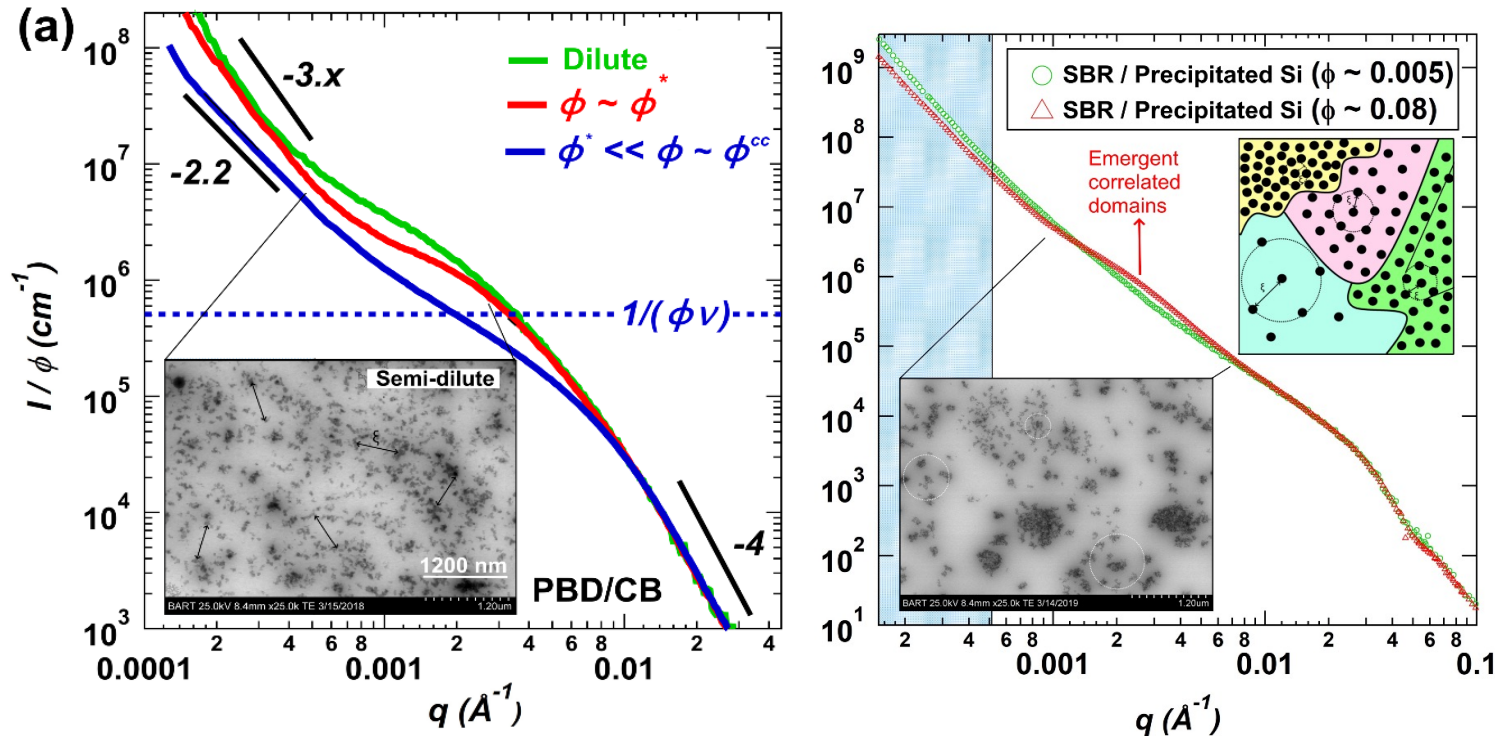
$$A_2 = 0, \quad t^* = a^*/b^*$$



N110	Vulcan 8 (Cabot)	123 m <sup>2</sup> /g	25.7 nm
N330	Vulcan 3 (Cabot)	76 m <sup>2</sup> /g	41.6 nm

Rishi, K.; Narayanan, V.; Beaucage, G.; McGlasson, A.; Kuppa, V.; Ilavsky, J.; Rackaitis, M. A Thermal Model to Describe Kinetic Dispersion in Rubber Nanocomposites: The Effect of Mixing Time on Dispersion. Polymer (Guildf). 2019, 175, 272–282. 4

# Mean field (CB) and specific interactions (Silica)



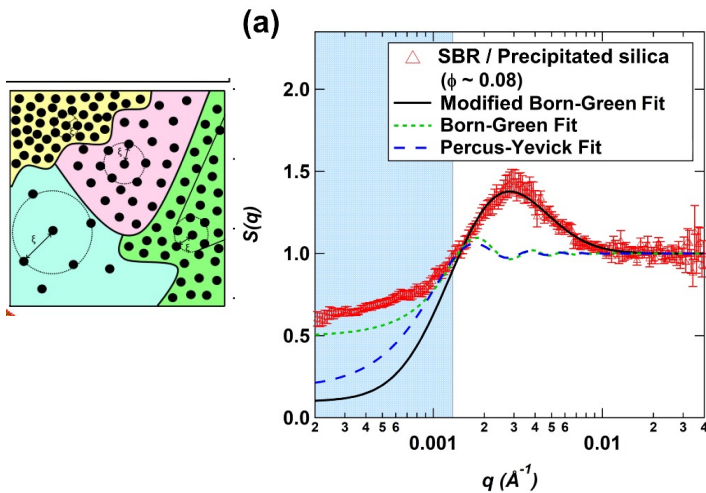
McGlasson, A., Rishi, K., Beaucage, G., Chauby, M., Kuppa, V., Ilavsky, J. and Rackaitis, M., 2020. Quantification of dispersion for weakly and strongly correlated nanofillers in polymer nanocomposites. *Macromolecules*, 53(6), pp.2235-2248.

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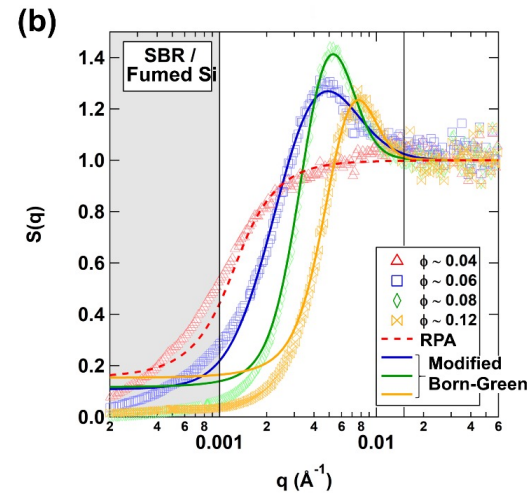
Rishi, K.; Pallerla, L.; Beaucage, G.; Tang, A. Dispersion of Surface-Modified, Aggregated, Fumed Silica in Polymer Nanocomposites. *J. Appl. Phys.* 2020, 127 (17), 174702.

# Specific interactions (Silica)

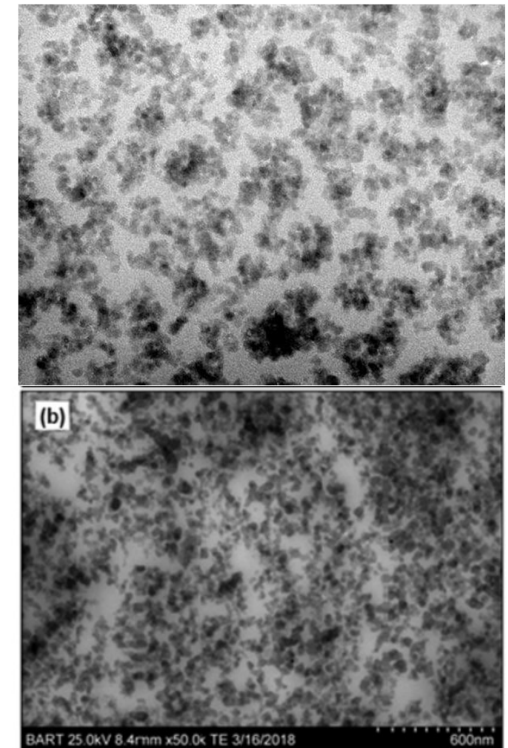
Positive  $a^*$  can lead to correlated silica aggregates, New scattering function to fit these curves that/ impossible task



Adapted from McGlasson, A., Rishi, K., Beaucage, G., Chauby, M., Kuppa, V., Ilavsky, J. & Rackaitis, M. Quantification of Dispersion for Weakly and Strongly Correlated Nanofillers in Polymer Nanocomposites. *Macromolecules* 53, 2235–2248 (2020).



Adapted from Rishi, K.; Pallerla, L.; Beaucage, G.; Tang, A. Dispersion of Surface-Modified, Aggregated, Fumed Silica in Polymer Nanocomposites. *J. Appl. Phys.* 2020, 127 (17), 174702.



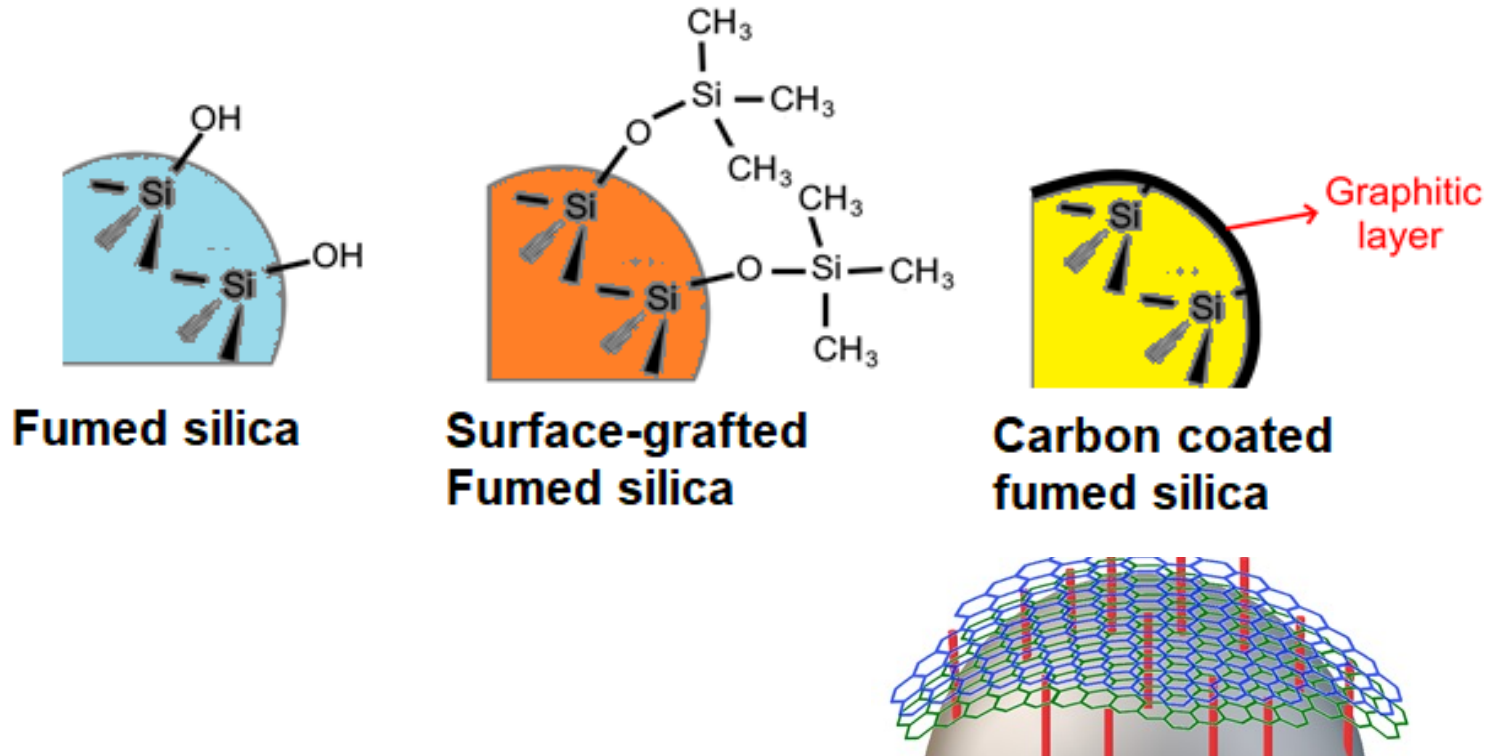
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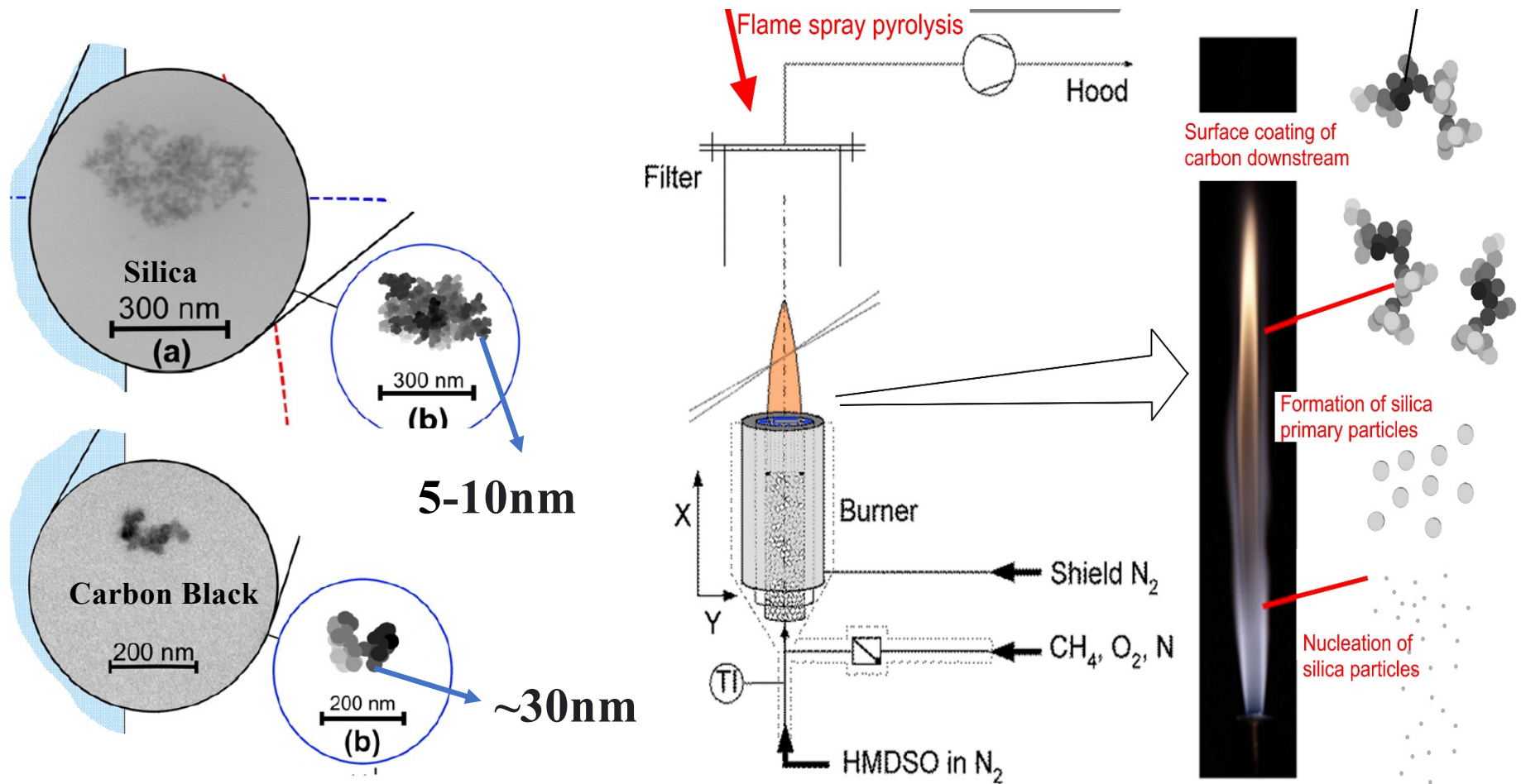


# Aggregates to clusters

Control immiscibility through surface modification



Okoli, U.; Rishi, K.; Beaucage, G.; Kammler, H. K.; McGlasson, A.; Michael, C.; Narayanan, V.; Grammens, J. *Dispersion and Dynamic Response for Flame-Synthesized and Chemically Modified Pyrogenic Silica in Rubber Nanocomposites*; 2021. In preparation.



McGlasson, A., Rishi, K., Beaucage, G., Chauby, M., Kuppa, V., Ilavsky, J. and Rackaitis, M., 2020. Quantification of dispersion for weakly and strongly correlated nanofillers in polymer nanocomposites. *Macromolecules*, 53(6), pp.2235-2248.

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# Surface Modification for Controlled Immiscibility

$$A_2(t) = \frac{N_A}{M^2} \left( b^* - \frac{a^*}{t} \right)$$

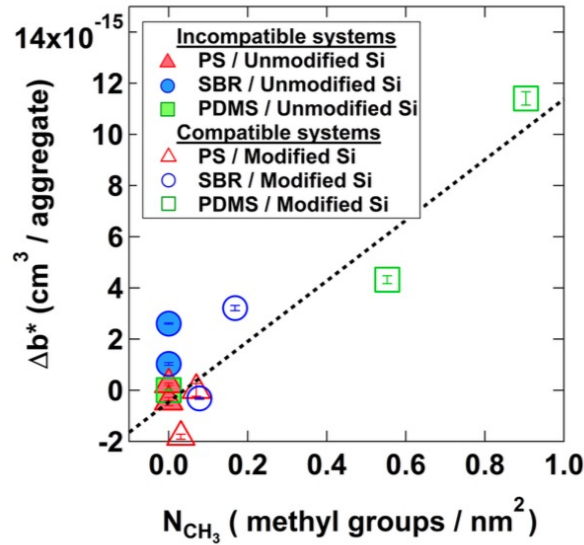
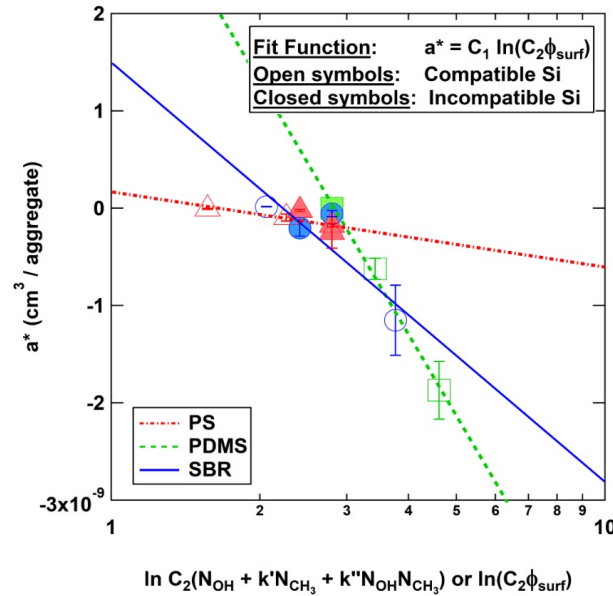


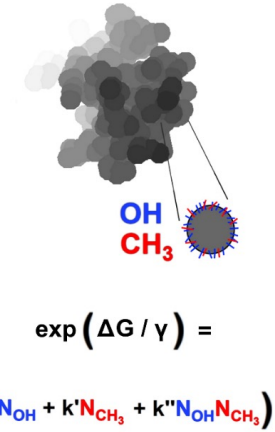
FIG. 11. A plot showing the bound polymer content ( $\Delta b^*$ ) determined from the excluded volumes of the filler aggregates before and after dispersion as a function of surface methyl content ( $N_{CH_3}$ ). The dashed line indicates that  $\Delta b^*$  is proportional to  $N_{CH_3}$  determined from FTIR.

$b^*$  can be calculated as the excluded volume for an aggregate,  $zV_0$ , without bound rubber

$b^*$  increases with bound rubber.



$a^*$  reflects the attractive energy of interaction between aggregates.  $\Pi = \frac{RT}{(V+b)} - \frac{a}{V^2}$ , attractive potential

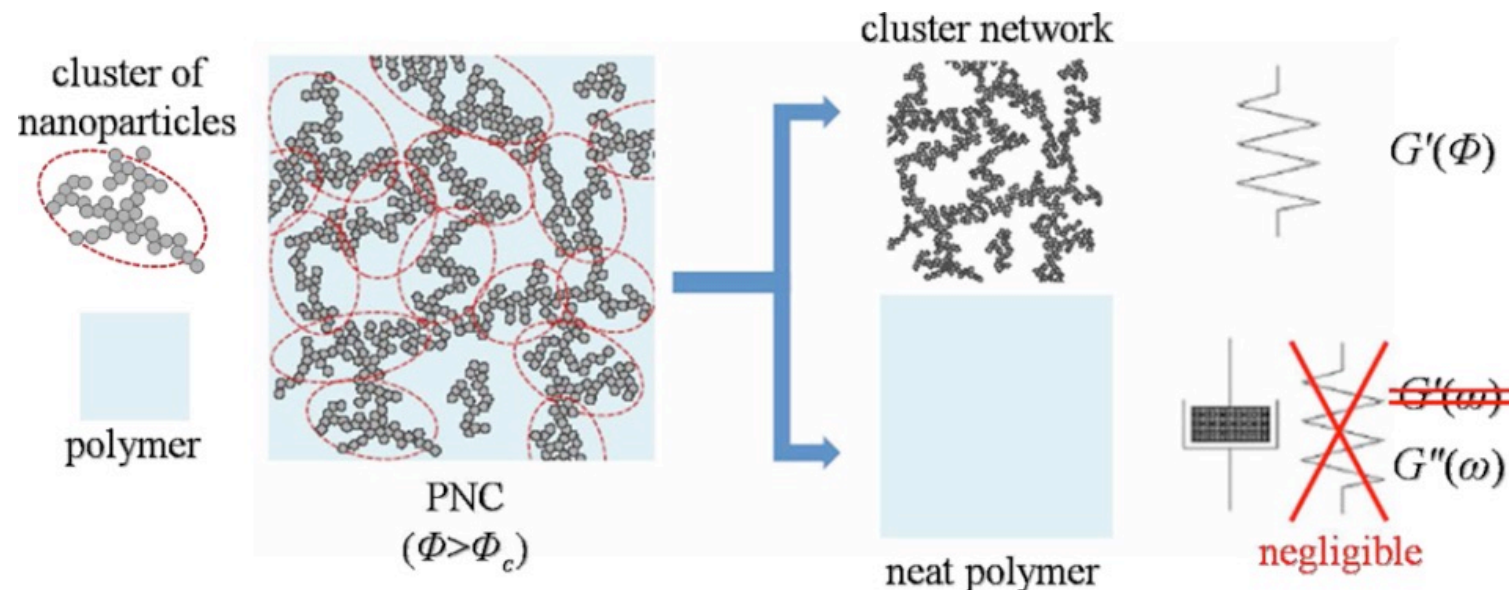


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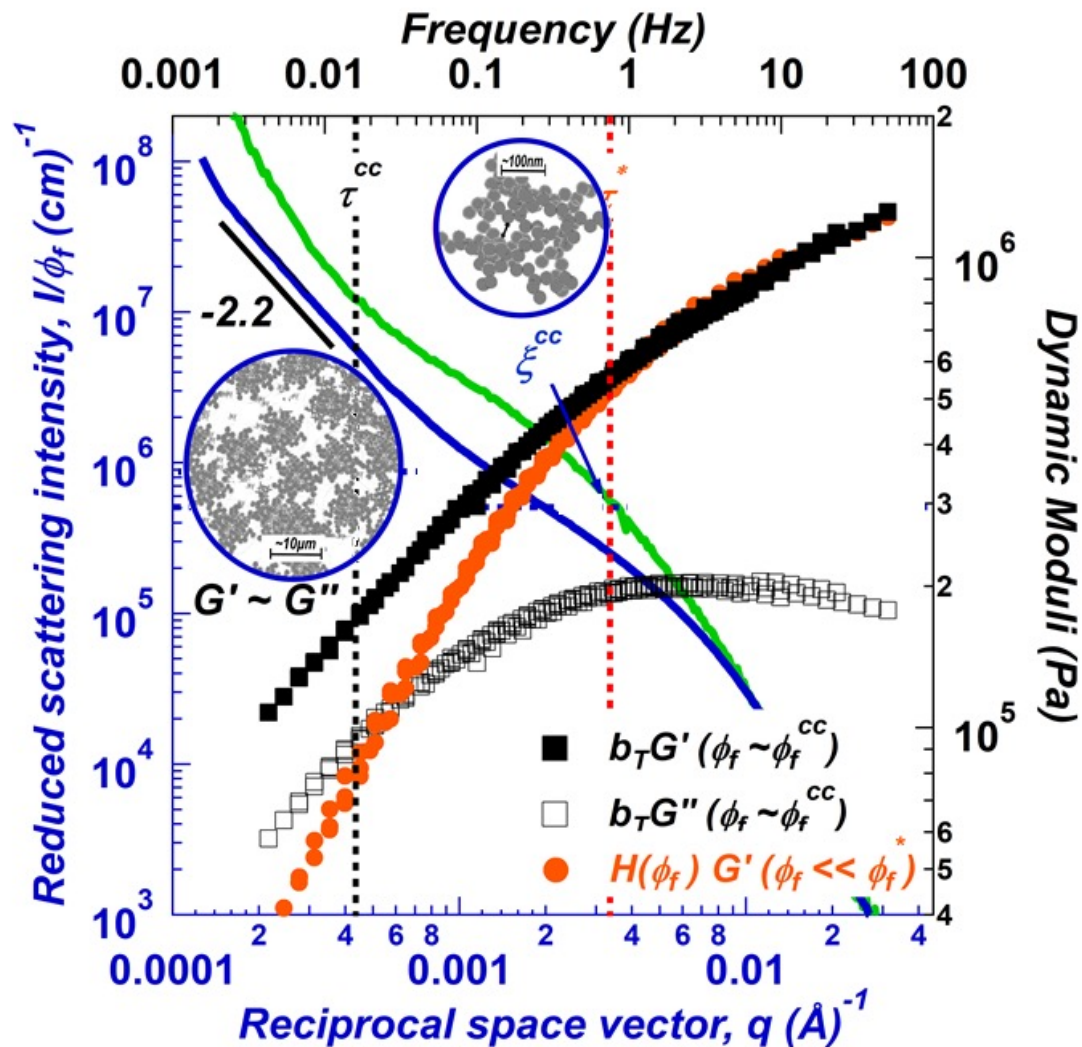
# Morphology from rheology

- How does this multi-hierarchical model relate to oscillatory rheometry?



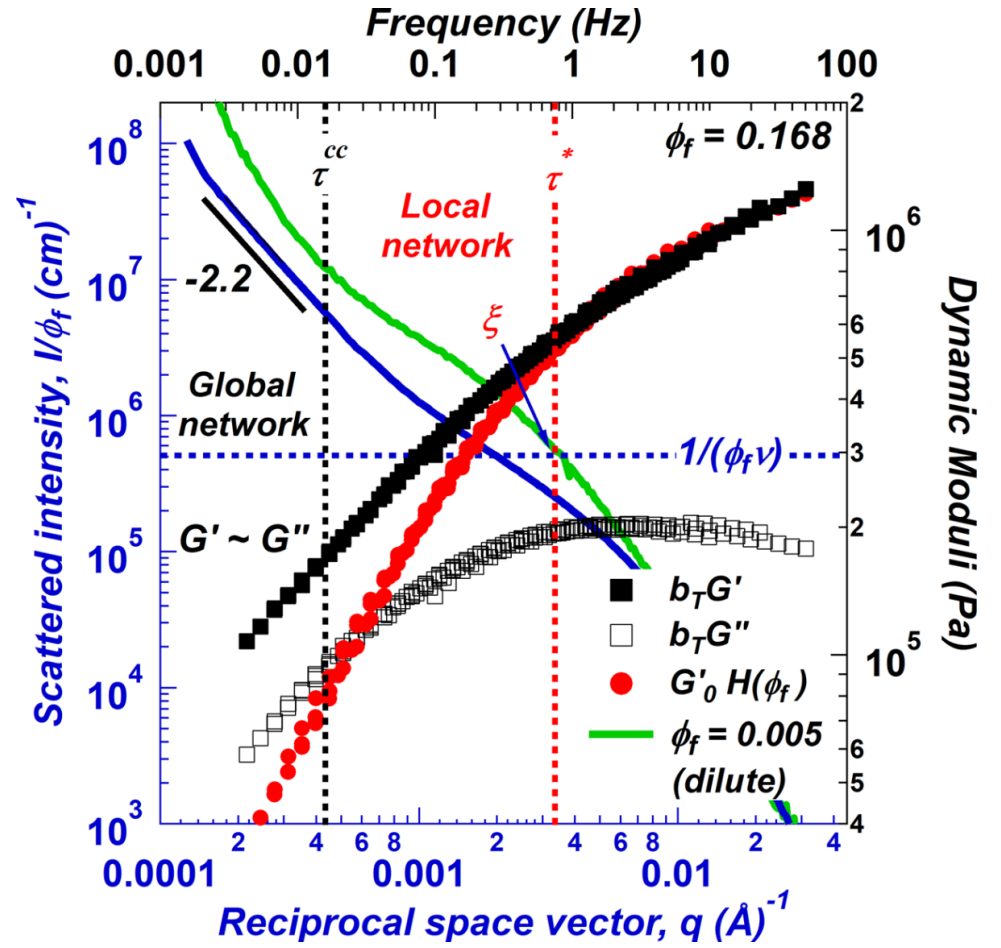
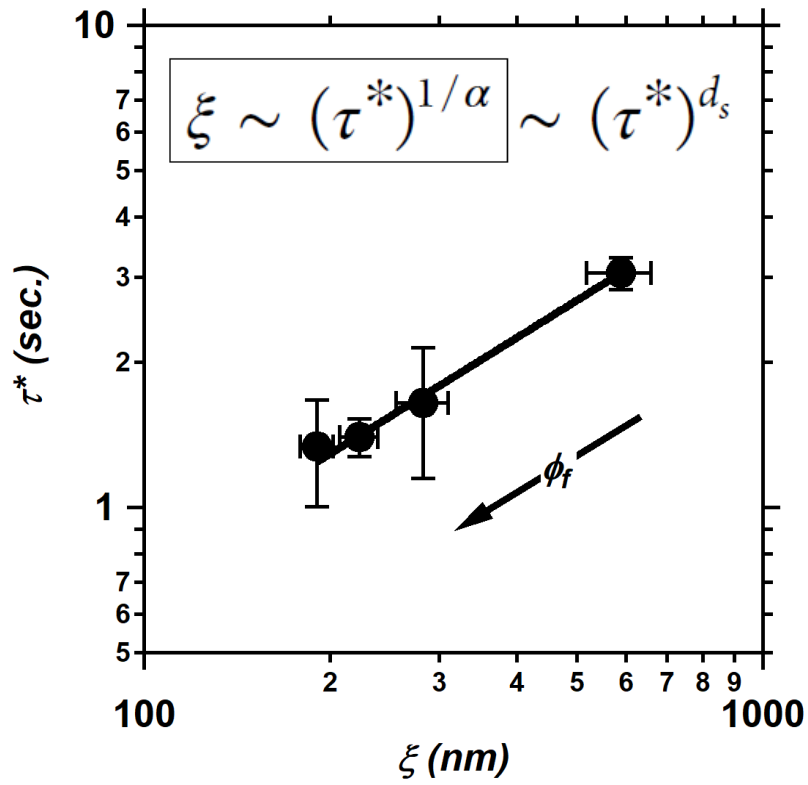
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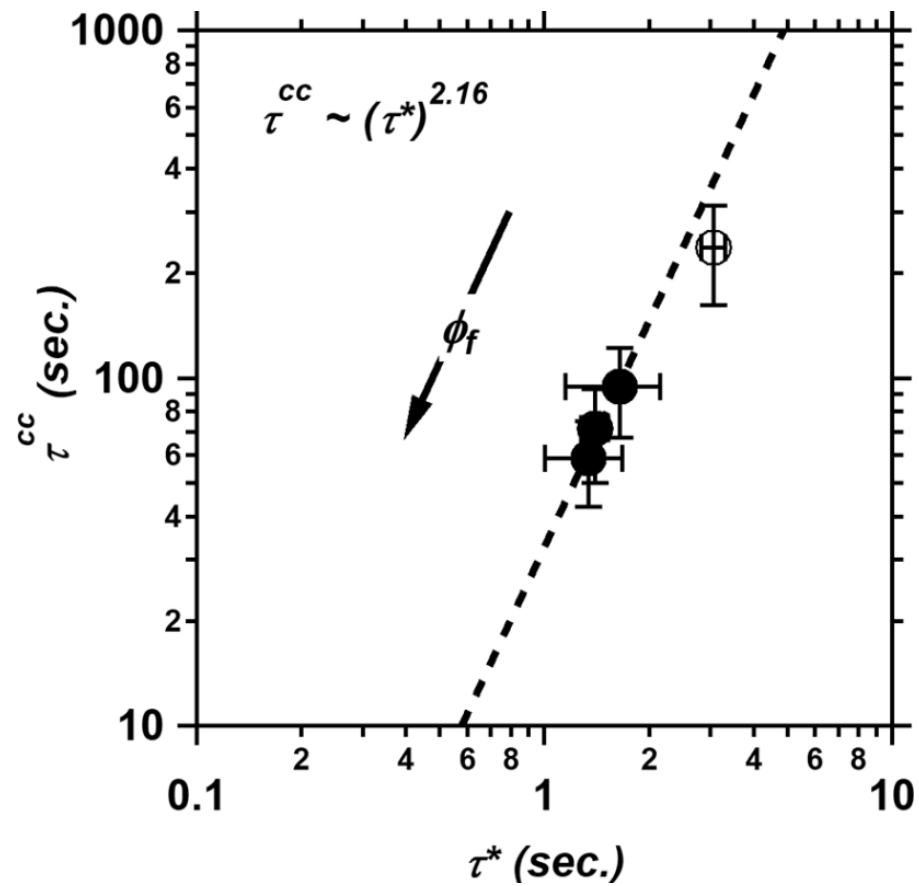
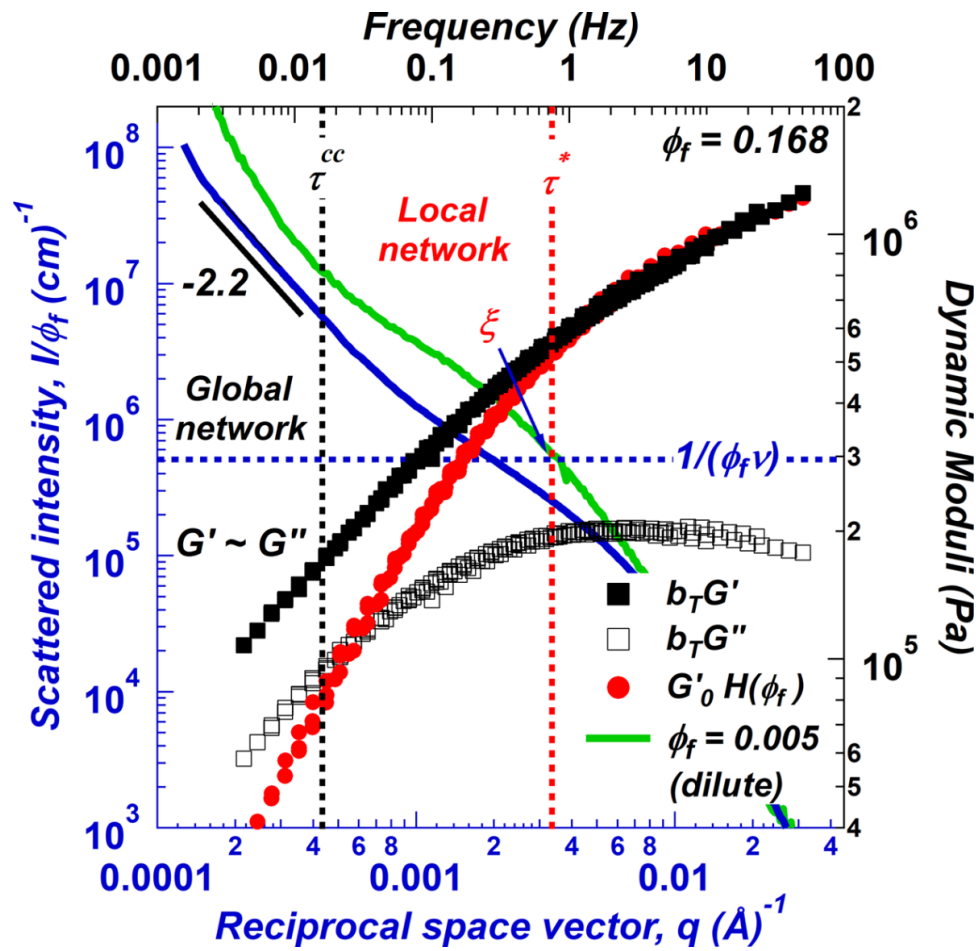


- ❖ At intermediate frequencies, deviation of semi-dilute rheology from dilute under same shear conditions ascertained by scaling dilute sample by Einstein-Smallwood factor
- ❖ At low oscillation frequencies,  $G' \sim G''$  indicates gel-like behavior\*



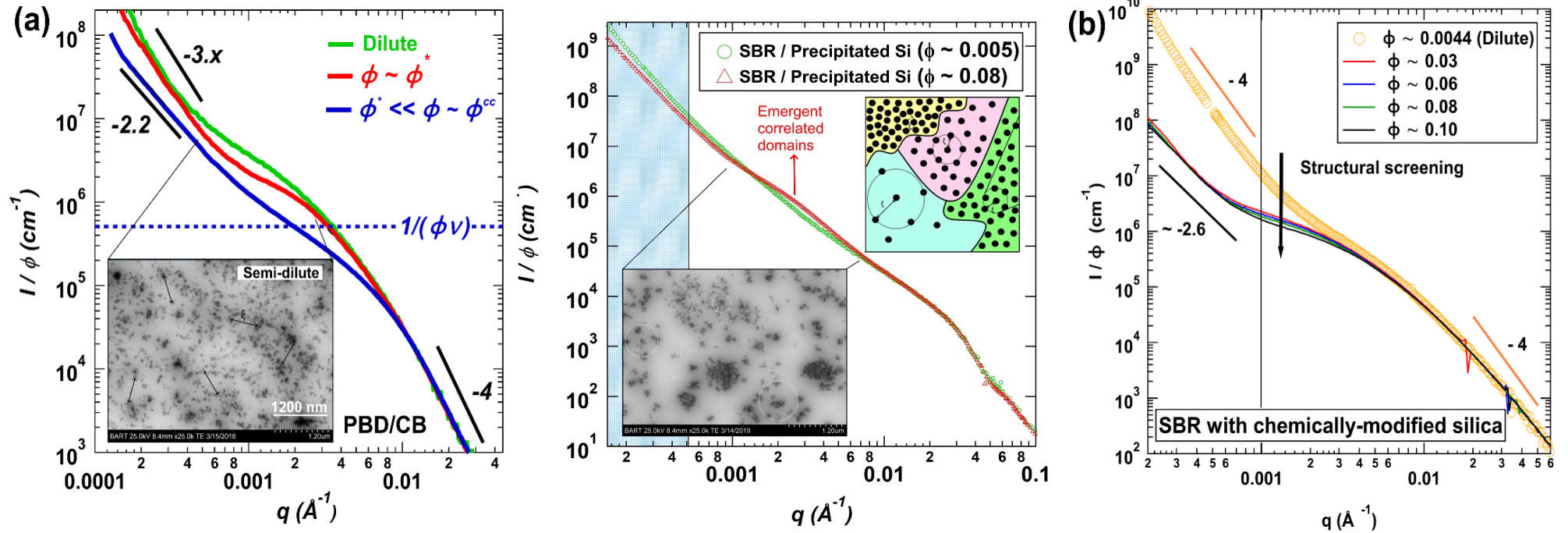


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# Clustered aggregates to bulk network



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# **Hierarchical emergent structure in commercial colloidal and polymeric systems**

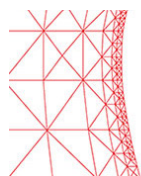
**Greg Beaucage**, Professor of Chemical and Materials Engineering

**Kabir Rishi**, PhD Candidate

Department of Materials Science & Engineering, University of Cincinnati

- There are multiple levels and not 1 that assemble from small to large sizes
- Primary particles decide aggregate structure, processing dictates changes to the aggregate structure and size
- Clustering of aggregates into local networks is dictated by the surface chemistry and kinetics.
- The formation of bulk network on the cm scale is dictated by the nature of clustered aggregates and the surface chemistry
- This constitutes a multiscale physics approach
- The point of compatibilization is to control immiscibility not to produce miscibility
- Link structure to dynamics in this complex system

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