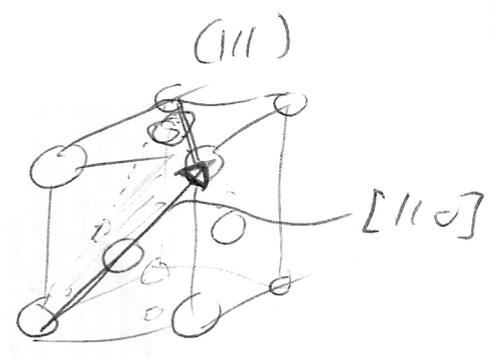


1) a)



b) This is the slip system for FCC.

c) We can calculate the yield stress for motion of an entire plane of atoms

$$\tau \sim \frac{G}{2\pi} \sim 3-30 \text{ GPa}$$

& this is 100 to 1000 times the observed yield stress

d)



Compression

The individual atom will tend to locate at the dislocation to relieve the tension. This makes motion of the dislocation more difficult

This leads to a higher yield stress  
& a harder material; e.g. steel vs. iron  
when the impurity is carbon in steel.

e) Point defects

    Vacancy

    Interstitial Atom (Carbon in steel)

    Substitutional Atom (Zn in Cu)  
    (Small or large)

    Frenkel defect

    Schottky defect

    Screw Dislocation

    Edge Dislocation

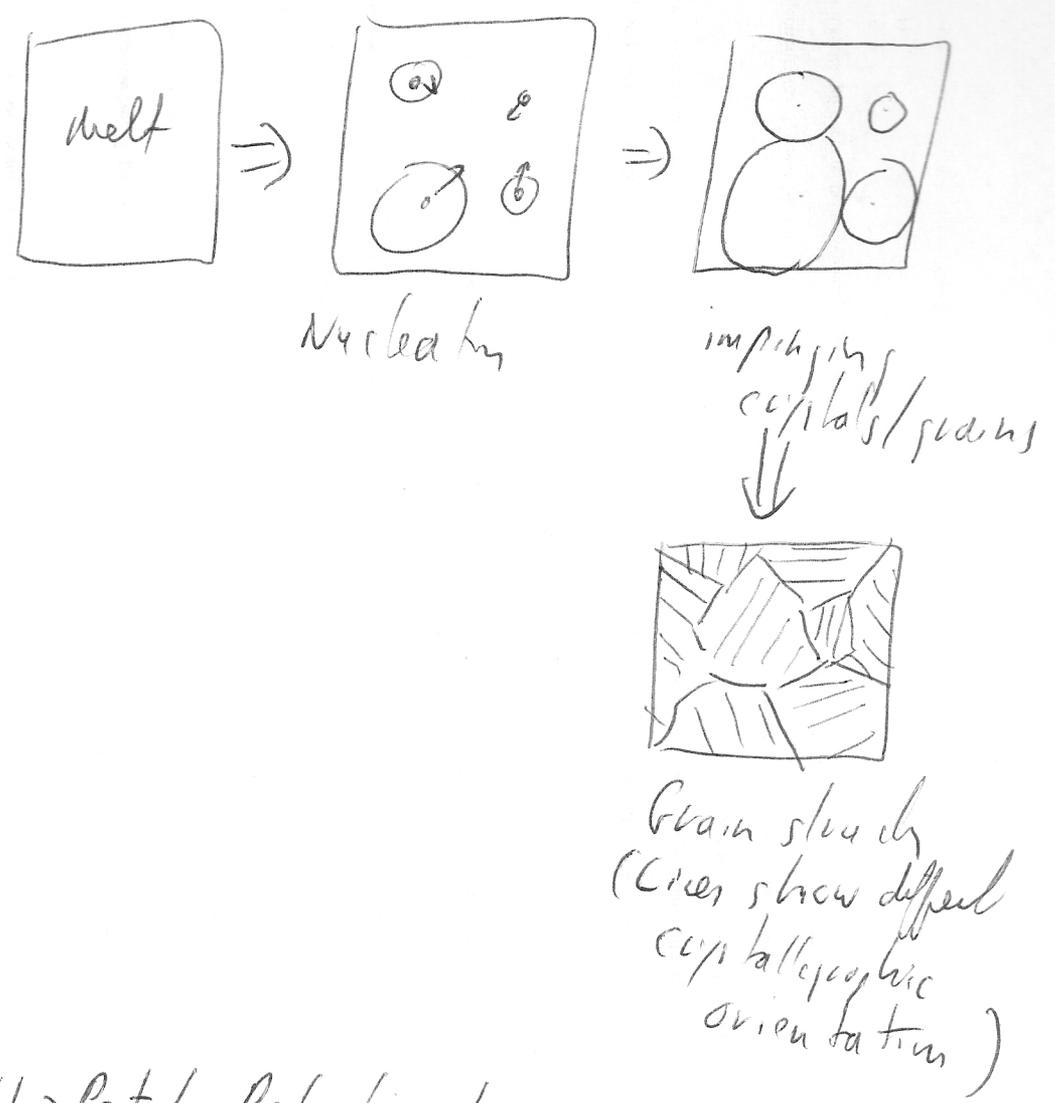
    Mixed Dislocation

    Twining surface defect

    Grain Boundary

        Small angle or large angle

2) a) Grains are single crystals in a polycrystalline sample. They form due to nucleation of different crystals during crystallization.



b) Hall-Petch Relationship

$$\sigma_{yield} = \sigma_i + \frac{k}{D^{1/2}}$$

$\sigma_i$   
No grains

$D = \langle \text{grain diameter} \rangle$

c) (i) each dot represent one grain in the x-ray beam. ~ 20 grains in a 20 μm beam so the grains are on the order of 3 μm in diameter.

(ii) Large grains give rise to streaks due to residual stress in grains

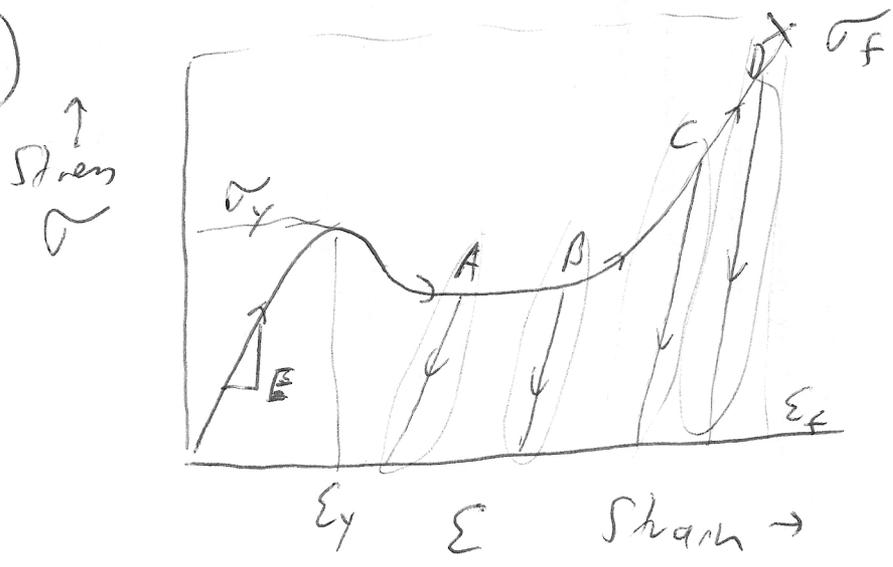
(iii) Large grains showing orientation of the crystals.

d) motion of dislocation has stop at the grain boundary.

e) A plane separates 2 twins. In FCC we have streaks of ABCABC in the [111] direction. A twin would occur if there were a shocky fault. ABCAB(A)BCABC  
Twin Plane

Other crystal systems can display twins.

3) a)



b) Modulus increases with strain harden  
 (E slope of  $\sigma$  vs  $\epsilon$ )  
 Yield Stress increases

c) 
$$\tau = \tau_0 + G \alpha b \rho^{1/2}$$
 No distance                      dislocation density  
 (wiki: Work Hardening)

d)e) see wiki page on strain hardening